

Convergence Issues

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Thanks to: D. Sánchez-Portal, E. Anglada, J.M. Soler, A. García, E. Artacho

You have to do **TESTS!**

Density functional theory

Different basis sets

Richard Martin's classification

1) Plane waves and grids:

Pros: Expansion in plane waves is simple and easy to implement.
Systematic way of improving the calculation

Cons: Large number of plane waves → memory requirements
Vacuum as expensive as matter

Codes: **Quantum Espresso**, VASP, Abinit, CASTEP...

2) Localized atomic-like orbitals: *gaussians, numerical atomic-like orbitals...*

Pros: Very efficient, no cost for vacuum

Cons: Harder to reach basis convergence, greater care in the election of the basis

Codes: **SIESTA**, Gaussian, CRYSTAL, OpenMX...

3) Atomic sphere methods: *FP-LAPW, LMTO, KKR...*

Atomic-like features near the nucleus, smooth functions between atoms

Pros: Very precise, best of both worlds

Cons: Computationally expensive, difficult to implement

Codes: **FLEUR**, Wien2k, RSPt, EXCITING...

If applied carefully, all of them agree after convergence

Convergence: precision vs accuracy

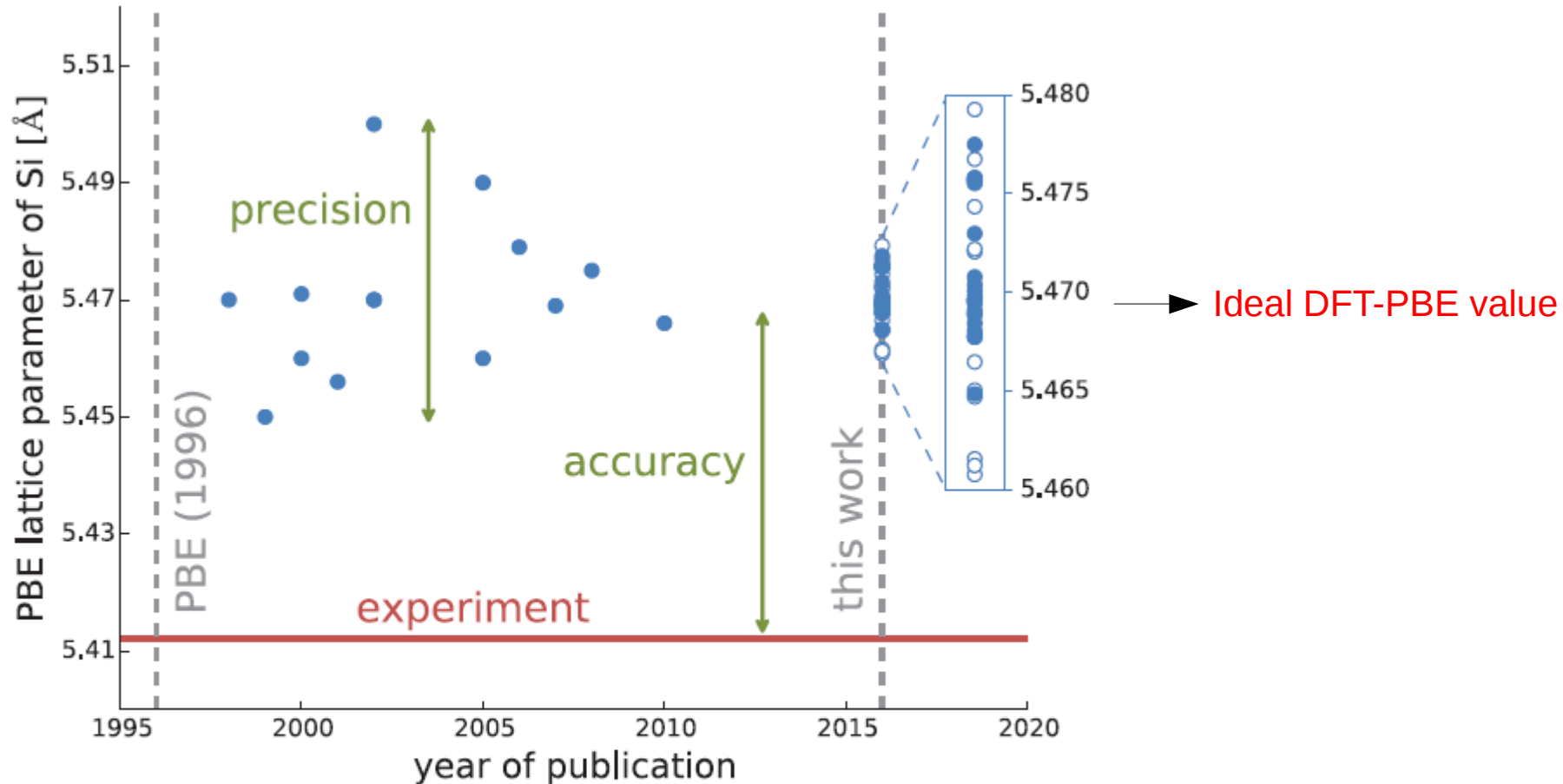


Fig. 1. Historical evolution of the predicted equilibrium lattice parameter for silicon. All data points represent calculations within the DFT-PBE framework. Values from literature (data points before 2016) (15, 16, 18, 56–62, 63–65) are compared with (i) predictions from the different codes used in this study (2016 data points, magnified in the inset; open circles indicate data produced by older methods or calculations with lower numerical settings) and (ii) the experimental value, extrapolated to 0 K and corrected for zero-point effects (red line) (26). The concepts of precision and accuracy are illustrated graphically.

Kurt Lejaeghere et al., “Reproducibility in density functional theory calculations of solids”. *Science* **351**, aad3000 (2016)

Precision and accuracy of the calculation

- DFT
 - XC functional
- } Accuracy
- Pseudopotentials
 - Basis set
- Reciprocal space grid: K-sampling
 - Real-space grid
 - Electronic temperature
 - ...
- } Precision

Basic strategy

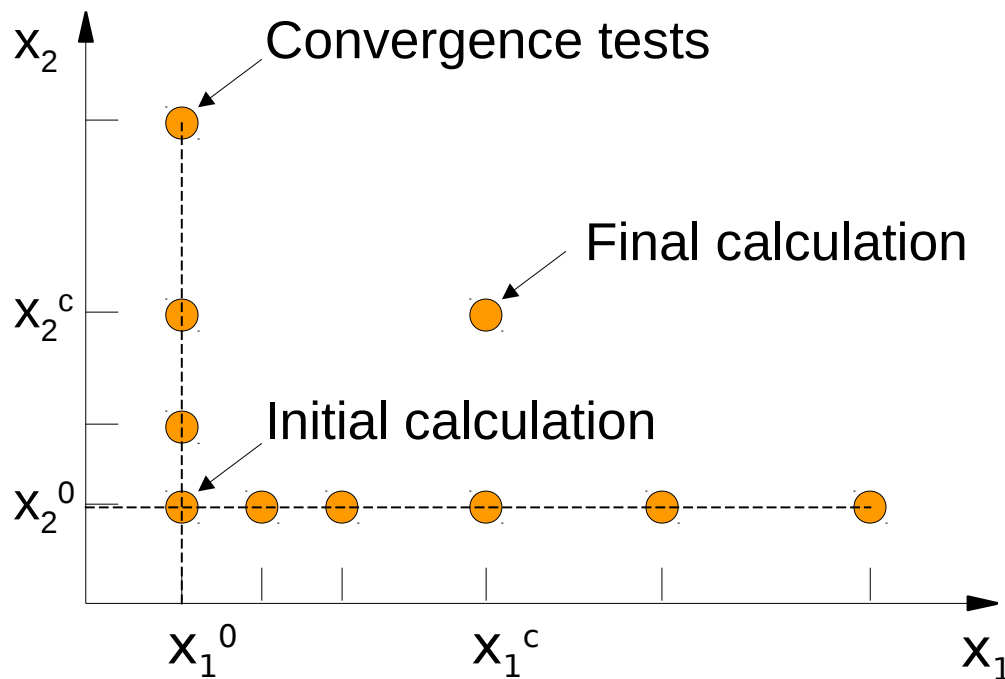
Typical steps in a research project:

- 1) Exploratory and feasibility tests
- 2) Convergence tests
- 3) Final converged calculations

It is impossible to obtain fully converged calculations
without convergence tests

Convergence tests

- Choose a relevant magnitude A of the problem (e.g. lattice constant, binding energy, magnetic moment...)
- Choose a set of parameters x_i (e.g. size of your basis set, number of k-points, grid mesh cutoff...)



Goal:

$$\frac{\delta A}{\delta x_i} < \text{our tolerance}$$

Monitor:

- Convergence
- CPU time
- Memory

Convergence tests

Practical hints

- Objective: qualitative or quantitative results?
- Do not try a converged calculation from the start
- Start with minimum values of all x_i
- Do not assume convergence for any x_i
- Choose a simpler reference system for some tests
- Refrain from stopping tests when results are “good”

Parameter list

- **XC functional:** LDA, GGA, vdW, LDA+U...
- **Pseudopotential:**
 - Method of generation
 - Number of valence states
 - Number of angular momenta
 - Core matching radii
 - Nonlinear core corrections
- **Basis sets:**
 - Shape
 - Number of functions: highest angular momentum, number of zetas
 - Range
- K-point sampling
- Real-space grid: mesh cutoff
- Self-consistent cycle tolerance
- Supercell size (system and vacuum size)
- Electronic temperature
- Geometry relaxation tolerance

Parameter interactions

$$\frac{\delta^2 A}{\delta x_i \delta x_j} \neq 0$$

Number of k-points:

- Supercell size
- Geometry
- Magnetic moments

Mesh cutoff:

- Pseudopotential
- Nonlinear core corrections
- Basis set
- GGA

Insight into relevant parameters

- **Pseudopotential**
- Basis set
- K-point sampling
- Real-space grid
- Self-consistent cycle

Pseudopotentials

SIESTA

- Norm-conserving pseudopotentials:
 - Typically Troullier-Martins
 - Soon ONCV → Better transferability [D.R. Hamann, *Phys Rev B* **88**, 085117 (2013)]
<https://code.launchpad.net/~siesta-pseudos-bases/siesta/trunk-psml> (advanced users)
- The choice of the pseudopotential is **critical** to obtain reliable results
 - Number of valence states
 - Which atomic reference configuration
 - Non linear core corrections?
 - Core radii
- Always generate pseudopotentials with great care
- Always **test** the pseudopotential in a well-known situation as close as possible to the systems you will be considering

Insight into relevant parameters

- Pseudopotential
- **Basis set**
- K-point sampling
- Real-space grid
- Self-consistent cycle

Basis sets

SIESTA

The only requirements:

1. $\phi_{Ilmn}(\vec{r}) = R_{Il}(|\vec{r}_I|)Y_{lm}(\hat{r}_I)$
2. Finite support (zero beyond a given cutoff radius, r_c)

They can be:

- As many as you want (both l-channels and z's)
- Of any (radial) shape
- Of any cutoff radius
- Centered anywhere (not necessarily on atoms)

Basis sets

Convergence of the basis set size

- **Single- ζ (minimal or SZ)**

One **single radial function** per angular momentum shell occupied in the free atom

Improving the quality:

- **Radial flexibility:**

Add more than one radial function within the same angular momentum than SZ

→ **Multiple- ζ**

+ Diffuse orbitals, floating orbitals...

- **Angular flexibility:**

Add shells of different atomic symmetry (different l)

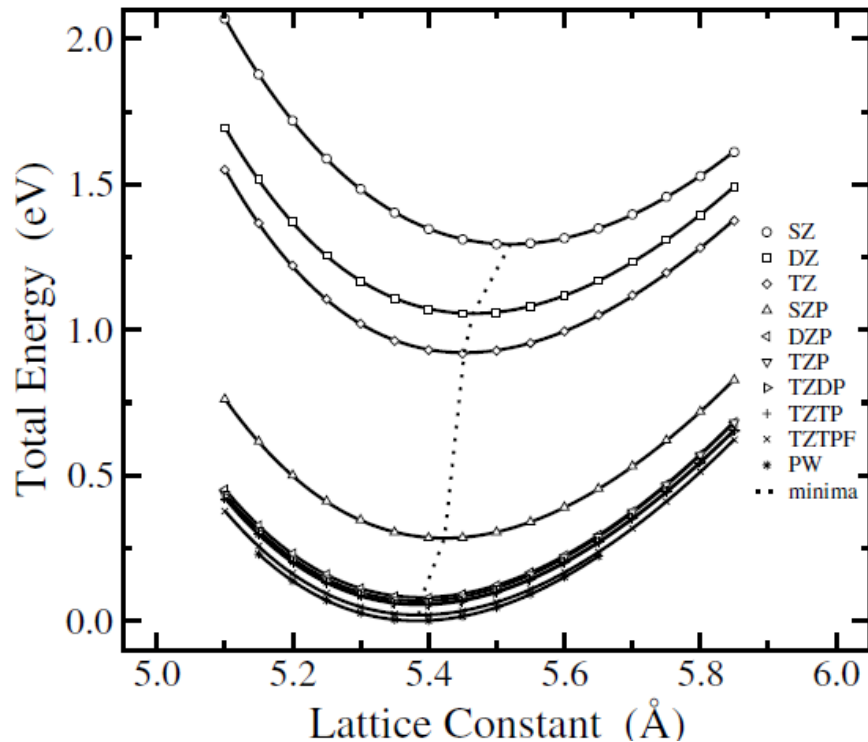
→ **Polarization**

Basis sets

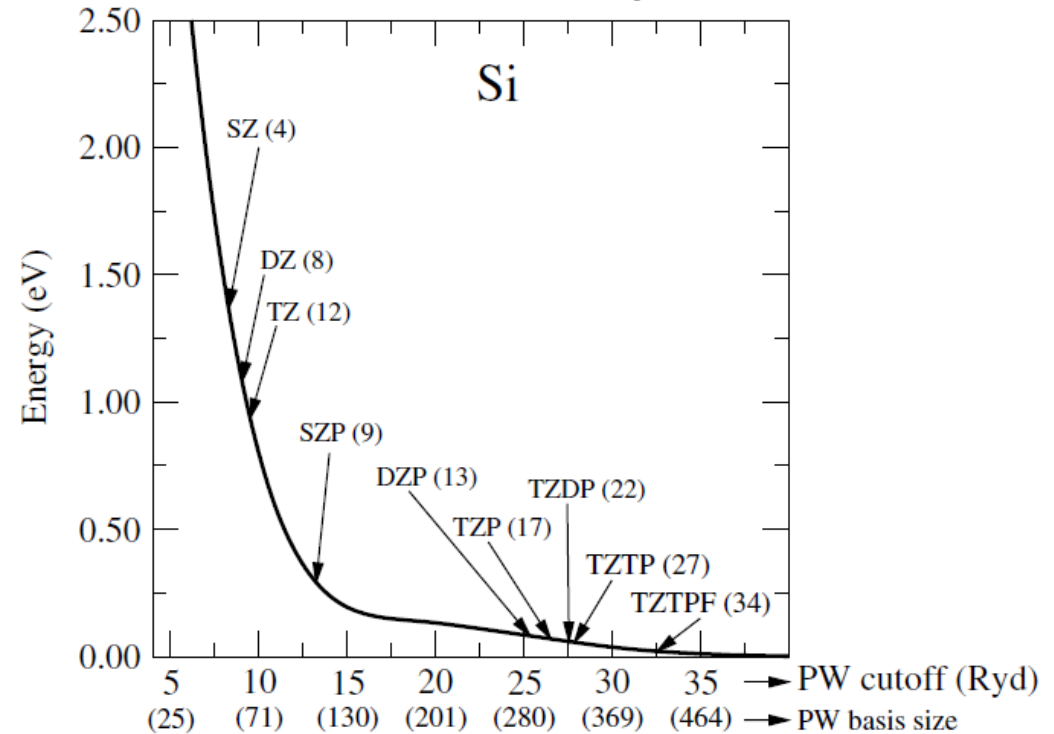
Convergence of the basis set size

Bulk Si

Cohesion curves



PW vs NAO convergence



J.M. Soler et al., "The SIESTA method for ab-initio order-N materials simulation". *J. Phys.: Condens. Matter.* **14**, 2745 (2002)

Basis sets

SZ basis

Bad results... but not as bad as you might expect:

- Energetics changes considerably; however, energy differences might be reasonable enough
- Charge transfer and other basic chemistry is usually OK
(*at least in simple systems*)
- If the geometry is set to the experiment, we typically have a nice band structure for occupied and lowest unoccupied bands

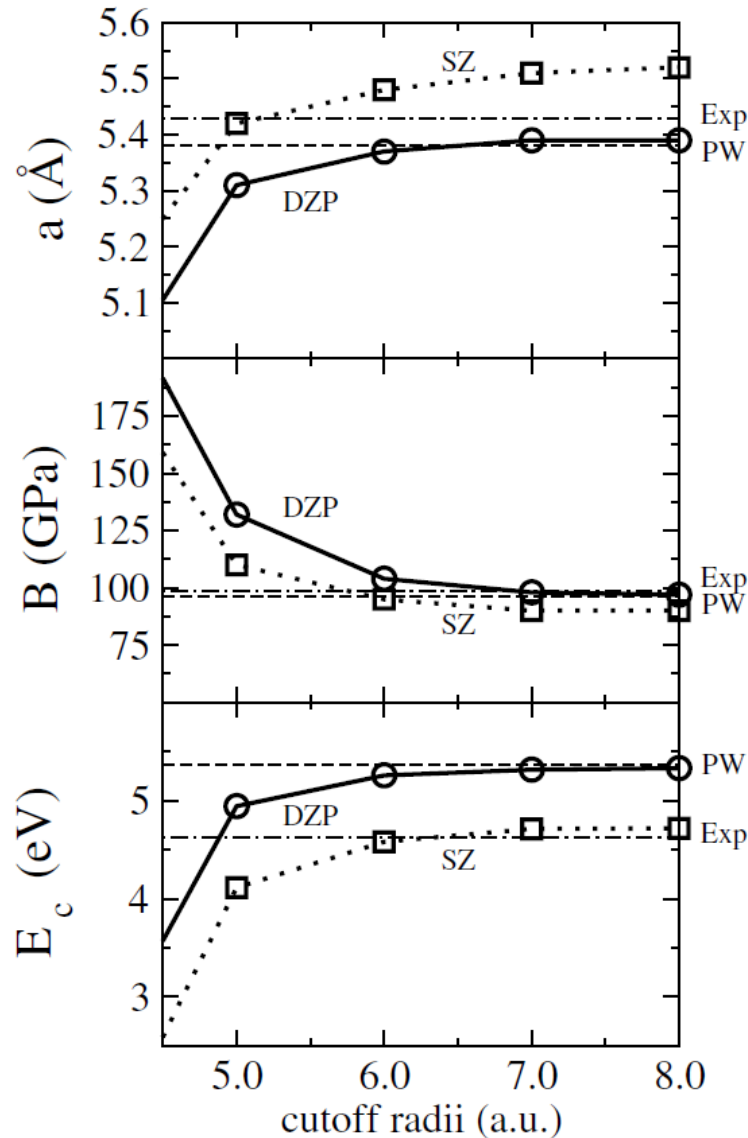
When can SZ basis sets be used:

- Long molecular dynamics simulations
(once we are sure that energetics is reasonable)
- Exploring very large systems and/or systems with many degrees of freedom
(complicated energy landscape)

Basis sets

Convergence with the range (r_c)

Bulk Si
equal s,p
orbital radii



Remarks:

- For incomplete bases (SZ), increasing radii may not produce better properties
- Affects total energies, but energy differences converge better
- More relevant for surfaces, small molecules, adsorbates
- Increasing the range makes the calculation more expensive!

J.M. Soler et al., "The SIESTA method for ab-initio order-N materials simulation". *J. Phys.: Condens. Matter.* **14**, 2745 (2002)

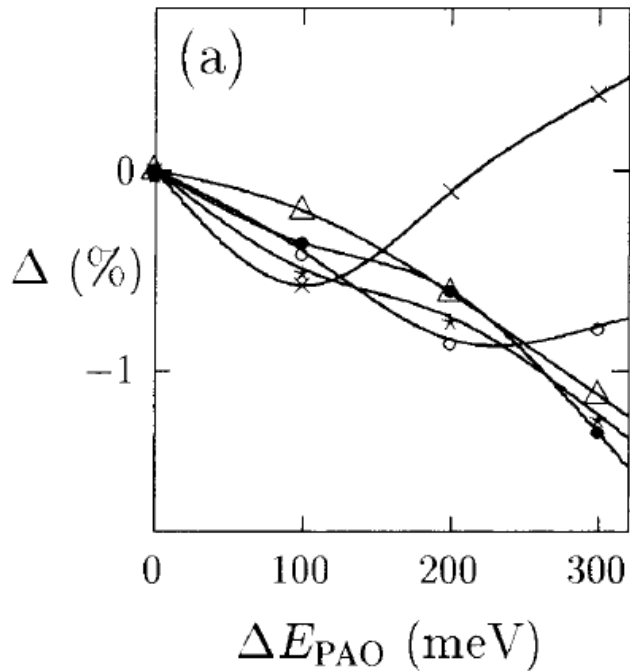
Basis sets

Range: energy shift

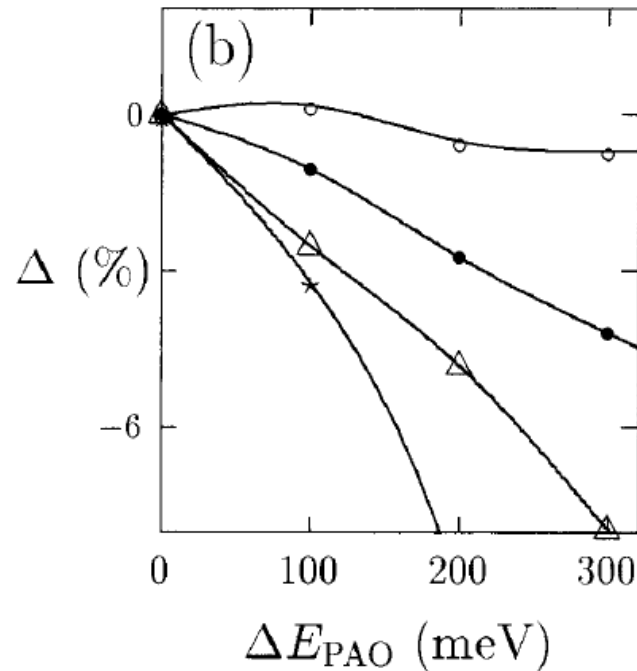
A single parameter for all cutoff radii $\rightarrow \Delta E_{\text{PAO}}$

Convergence vs Energy shift of

Bond lengths



Bond energies



Reasonable values
for practical calculations
 $\Delta E_{\text{PAO}} \sim 50-200$ meV

E. Artacho et al., *Phys. Stat. Solidi (b)* **215**, 809 (1999)

Basis sets

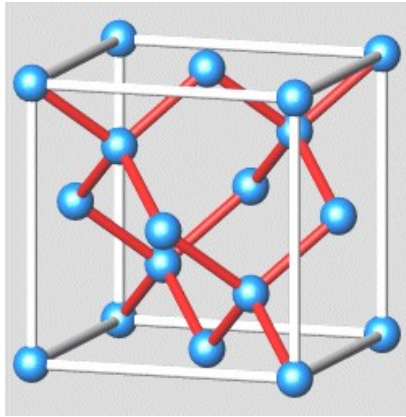
Procedure

- **SZ:** (energy shift)
Semiquantitative results and general trends
- **DZP:** automatically generated (energy shift, split valence)
Good quality for most systems
Good balance: converged results vs computational cost
Default, but **tests are always needed**
- **Variational optimization:**
Diffuse orbitals
Floating orbitals
→ Example: noble metal surfaces (S.García-Gil, *Phys. Rev B* **79**, 075441 (2009))

Insight into relevant parameters

- Pseudopotential
- Basis set
- **K-point sampling**
- Real-space grid
- Self-consistent cycle

K-point sampling



$$V(\mathbf{r}) = V(\mathbf{r} + \mathbf{R})$$

Bloch Theorem: The eigenstates of the one-electron Hamiltonian in a periodic potential can be chosen to have the form of a plane wave times a function with the periodicity of the Bravais lattice.

$$\psi_{n,\mathbf{k}}(\mathbf{r}) = e^{i\mathbf{k}\cdot\mathbf{r}} u_{n,\mathbf{k}}(\mathbf{r})$$

$$u_{n,\mathbf{k}}(\mathbf{r} + \mathbf{R}) = u_{n,\mathbf{k}}(\mathbf{r})$$

Periodicity in reciprocal space
 $\mathbf{G} \equiv$ Reciprocal lattice vector

$$\begin{cases} \psi_{n,\mathbf{k}+\mathbf{G}}(\mathbf{r}) = \psi_{n,\mathbf{k}}(\mathbf{r}) \\ \varepsilon_{n,\mathbf{k}+\mathbf{G}} = \varepsilon_{n,\mathbf{k}} \end{cases}$$

Instead of computing an infinite number of
electronic wave functions



Finite number of wave functions at an infinite
number of k-points in the 1BZ

K-point sampling

Many quantities require integration of Bloch functions over the first Brillouin zone (BZ):

$$n(\mathbf{r}) = \sum_i^{\text{occ}} \int_{BZ} |\psi_{i\mathbf{k}}(\mathbf{r})|^2 d\mathbf{k}$$

$$E_{BS} = \sum_i \int_{BZ} n_i(\mathbf{k}) \varepsilon_i(\mathbf{k}) d\mathbf{k}$$

In principle: we should know the eigenvalues and/or eigenvectors at all the k-points in the first BZ

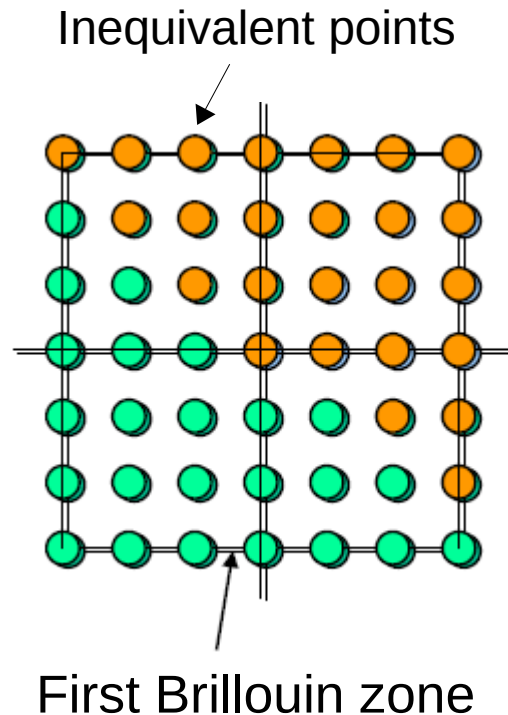
In practice: electronic wave functions at k-points that are very close together will be almost identical

→ It is possible to represent electronic wave functions over a region of reciprocal space by the wave function at a single k-point

→ Approximated by sums over selected k-points

$$\int d\mathbf{k} \longrightarrow \sum_{\mathbf{k}} \Delta\mathbf{k}$$

K-point sampling

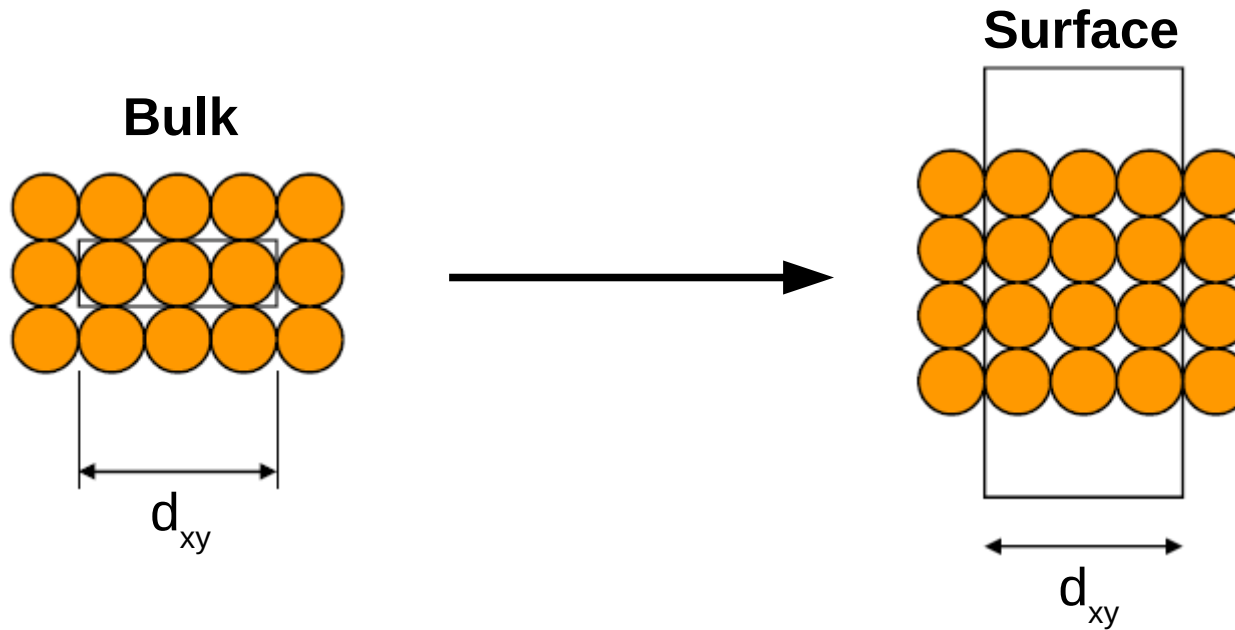


- Controlled by *kgrid.Cutoff* or *kgrid.MonkhorstPack*
- Only time reversal symmetry is used in SIESTA
- Convergence is equal to other codes:
 - Metals require a lot of k-points
 - Insulators require less k-points
- An incremental procedure is the best approach:
 - Density matrix and geometry are good with a “reasonable” number of k-points
 - Use a “reasonable” number for most calculations, then increase the number to refine the results

K-point sampling

The BZ sampling must be similar for all the considered structures:
→ Try to keep the same density of k-points

Example: surface (slab) calculations



For the same d_{xy} use the same K_{xy} points

Insight into relevant parameters

- Pseudopotential
- Basis set
- K-point sampling
- **Real-space grid**
- Self-consistent cycle

Real-space grid

It is used to project $\rho(\mathbf{r})$ in order to calculate:

- XC potential
- Solve Poisson equation to get Hartree potential
- Calculate three center integrals (difficult to tabulate and store)

The real-space grid is **not** part of the basis set, is an auxiliary grid and, therefore, convergence of energy is not necessarily variational with respect to its fineness.

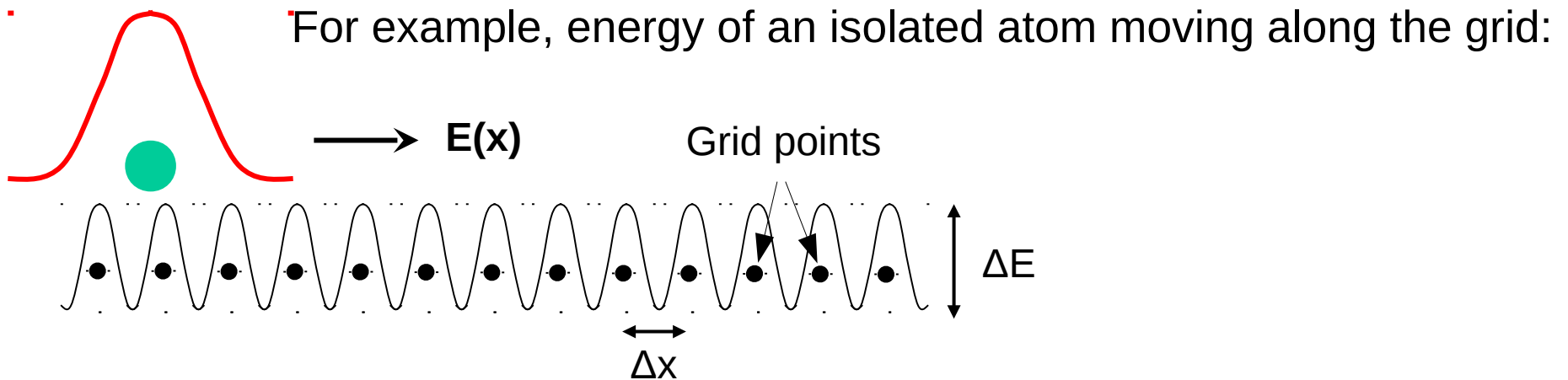
Mesh cutoff

- Single parameter which controls the fineness of the grid
- It is the maximum kinetic energy of the plane waves that can be represented in the grid without aliasing

Real-space grid

“Eggbox” effect

The real-space grid breaks translational symmetry: “eggbox” effect



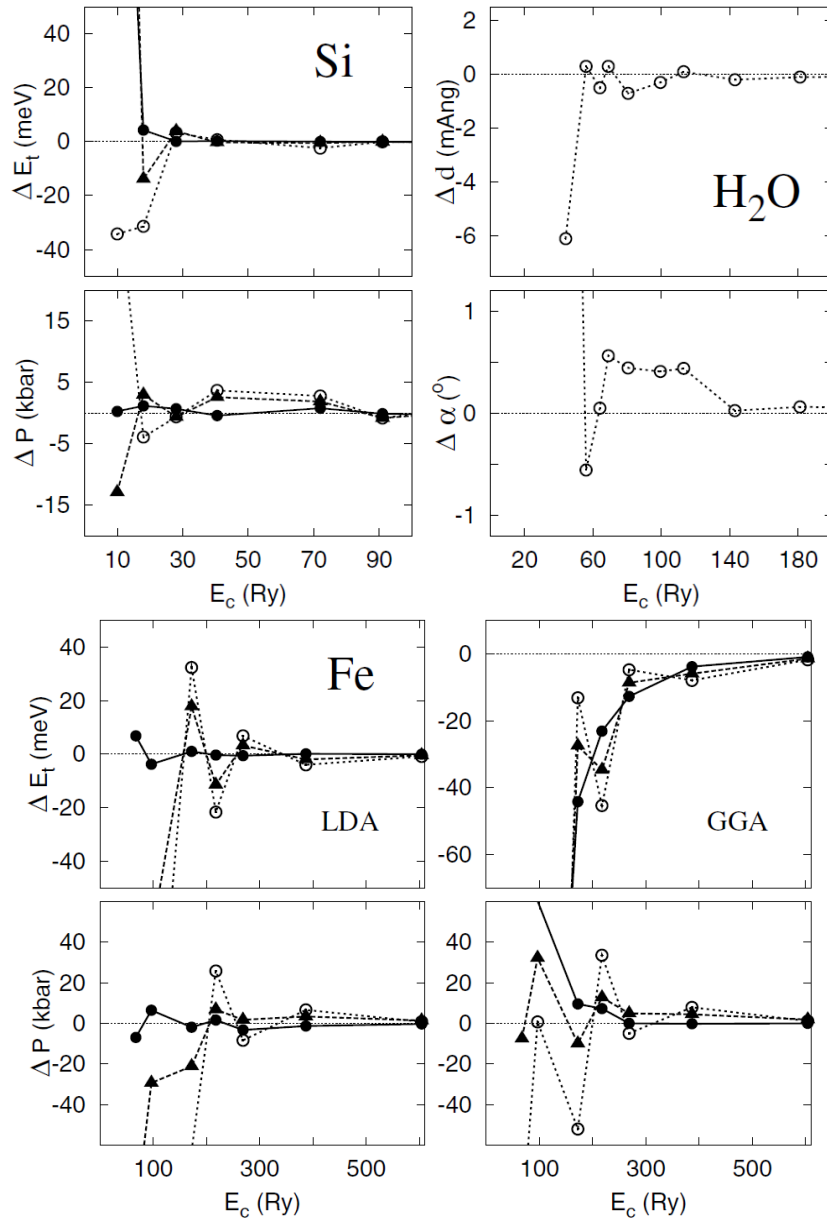
- Forces are more affected than energy
- Important for very precise relaxations and phonon calculations

Solutions:

- Increase Mesh cutoff
- *Grid.CellSampling*
- Filtering of atomic orbitals and potentials
(E. Anglada et al, *Phys. Rev B* **73**, 115122 (2006))

Real-space grid

Mesh cutoff convergence

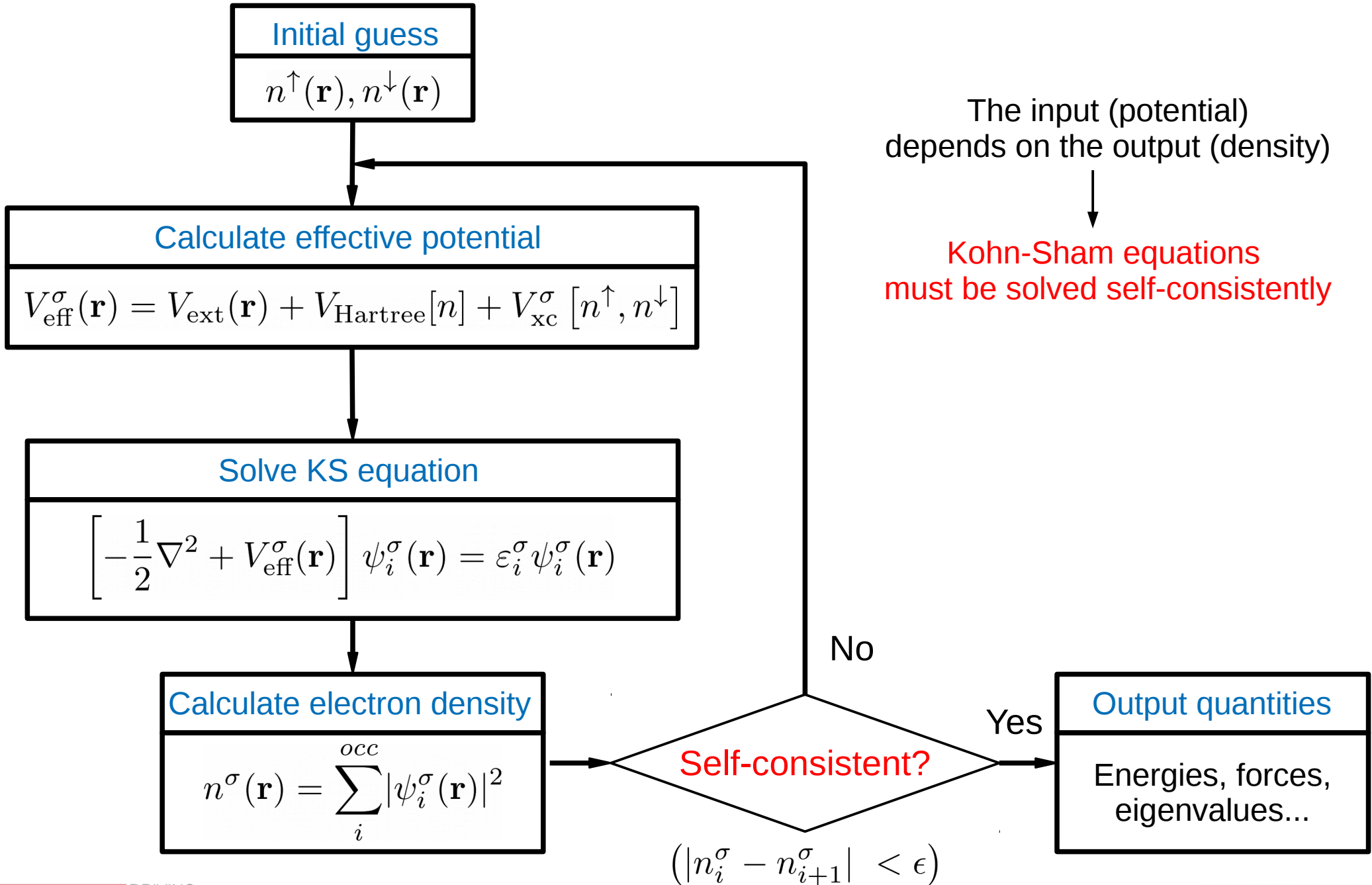


- Convergence is rarely achieved below 100 Ry
- Starting value around 150-200 Ry
- GGA and non-linear core corrections require higher values
- Options to improve:
 - *Grid.CellSampling*
 - Filtering of orbitals and potentials (E. Anglada et al, *Phys. Rev B* **73**, 115122 (2006))

Insight into relevant parameters

- Pseudopotential
- Basis set
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- **Self-consistent cycle**

Self-consistent cycle



Self-consistent cycle

“How much do I need to converge?”: value of ϵ

$$|n_i^\sigma - n_{i+1}^\sigma| < \epsilon$$

- Density matrix as monitor for self-consistency:
SCF.DM.Tolerance (default: 10^{-4} eV)
 - Stick to this value or smaller
 - Higher values could be considered in special situations, like preliminary relaxations
 - Above 10^{-3} eV the errors are too large
- Hamiltonian matrix elements as monitor for self-consistency:
SCF.H.Tolerance (default: 10^{-3} eV)
 - Default value is good in most situations
 - Could be difficult to achieve in some circumstances, higher values can be acceptable if DM is well converged
- Requirements depend on which properties we are interested in
- **Always check that things make sense**

Self-consistent cycle

Mixing

- Mixing of input and output quantities is needed to achieve self-consistency

$$\rho_{\text{in}}^{n+1} = \alpha \rho_{\text{out}}^n + (1 - \alpha) \rho_{\text{in}}^n$$

- We can mix:
 - Hamiltonian elements (default, usually faster)
 - Density matrix (old default)
- Many different options, only by testing you can find good parameters for your system:
 - *DM.MixingWeight*: α is system dependent and difficult to guess: 0.1-0.15 for insulators and semiconductors, smaller for metals
 - Type of mixing: linear, Pulay, Broyden
 - *DM.NumberPulay*: 3-7
- Self-consistency can be quite difficult to achieve for some systems
 - Magnetic systems (specially non-collinear)
 - Spin-orbit coupling

Conclusions

- We have different choices to make and parameters to determine, every one has to be carefully checked:
 - XC functional
 - Pseudopotential
 - Basis set
 - K-points
 - Grid cutoff
 - ...
- The best way to handle it is an incremental approach: start from a reasonable guess and converge the parameters until the desired precision is reached
- All the parameters are system dependent, everything has to be tested

TEST, TEST, TEST

You have to do **TESTS!**